Commensurate intergrowths in titanium monoboride precipitates

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The addition of boron to Ti-based and TiAl-based alloys can refine the grain structure and this is usually accompanied by the formation of boride particles due to the low solid solubility of B. The borides can exhibit a remarkable diversity of morphologies and crystal structures, although most of these are TiB monoborides in the form of plates, needles, and/or ribbons. The equilibrium crystal structure for TiB is B27, but certain alloying additions (e.g. Nb, Ta) can stabilize the B_f structure [1]. Such borides often contain high densities of planar defects and in our work we have studied these defect structures in two alloys: Ti-6-4 and Ti-44Al-4Nb-4Zr (at. %), both modified with 1 at.% B.

The alloys were prepared by transferred-arc plasma melting with B in the form of AlB_{12} . TEM samples were prepared by twin-jet electropolishing using an electrolyte consisting of 6% perchloric acid, 34% butanol and 60% methanol at -15°C and 30 V / 40 mA. The samples were then Ar+ ion milled for 0.5 h at 0.5 kV / 0.2 mA to remove any residual contamination or damage layer on the surfaces. The CTEM experiments were performed in a Philips EM420T operating at 120 kV whereas HRTEM lattice images were acquired using JEOL 2010 FasTEM equipped with a UHR objective lens polepiece ($C_s \approx 0.5$ mm) and operated at 200 kV.

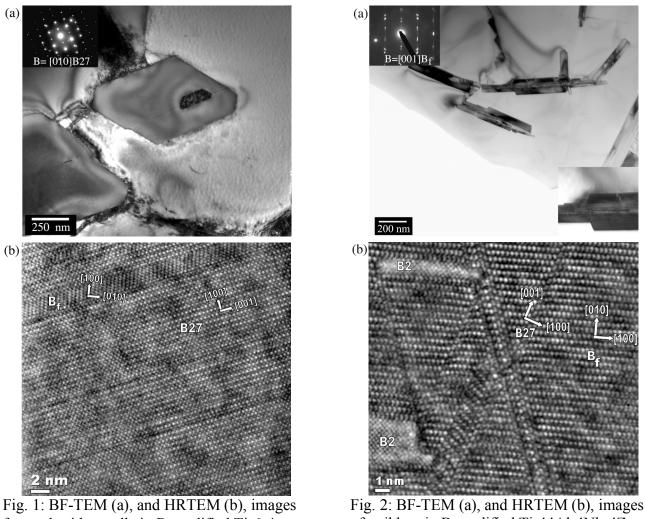
Preliminary SEM observations showed that the borides in B-modified Ti-6-4 were needle-like, whereas those in the B-modified Ti-44Al-4Nb-4Zr were curved ribbons. BF-TEM images (e.g. Fig. 1(a) & 2(a)) revealed that in both cases the borides contained planar faults parallel to the major facet. The needles in the B-modified Ti-6-4 consist of the B27-TiB phase and are oriented with [010] parallel to the needle axis (Fig. 1(a)). HRTEM images such as Fig. 1(b) reveal that the faults lie on (100), and that these correspond to thin coherent intergrowths of the B_f-TiB phase. The ribbons in the B-modified Ti-44Al-4Nb-4Zr are comprised of thin blocks with the B_f-TiB structure and having (010) as the major facet. These blocks contain a wide variety of different defect structures including pockets of metallic B2 phase, and coherent intergrowths of Ti₃B₄ and TiB₂: all of these features lie parallel to (010)B_f and the details of these structures are presented elsewhere [2]. Occasional faults were also observed lying parallel to (110) plane of B_f, and two examples are shown in Fig. 2(b). These latter faults correspond to thin embedded slabs of B27-TiB phase. Interestingly, in both the needles and the ribbons the orientation relationships between the B_f-TiB and B27-TiB phases is:

 $[001]B_f//[010]B27$; with $\{110\}B_f//(100)$ B27.

The B_f - and B27-TiB structures are closely related, both comprising regular trigonal prisms with six Ti atoms at the vertices and one B atom at the center [1,2]. Although these prisms are arranged rather differently in the two structures, it is possible to produce the B27-TiB structure by repeated 180° rotational twinning of the B_f structure on $\{110\}$ (Figure 3). Similarly, 180° rotational twinning of the B27 structure on (100) would produce B_f . As such, one would expect fully coherent interfaces, which would explain why the faults form on these planes. However, the faults are highly unlikely to form by a shear twinning mechanism in either of the alloy systems considered here: the orientation of the faults parallel to pronounced facets on the boride/metal interfaces suggests that they are instead probably the result of errors in the stacking sequence during growth.

References

- [1] M. De Graef, J. P. A. Lofvander and C. G. Levi. Acta Metall Mater 1991; 39: 2238
- [2] U.Kitkamthorn, L. C. Zhang and M. Aindow. Intermetallics 2006: In press
- [3] Work supported by NSF (Grant # DMR DMR-0072721) and by the Royal Thai Government.



from a boride needle in B-modified Ti-6-4.

of a ribbon in B-modified Ti-44Al-4Nb-4Zr.

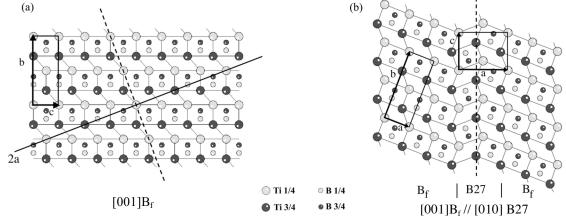


Fig. 3: Creation of a thin B27 layer at a 180° twin on (110) in the B_f structure [1]