Probing the Polar Instability of Strained SrTiO₃ with HAADF-STEM

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SrTiO₃ is an incipient ferroelectric in unstrained, pure form, but easily becomes ferroelectric when subjected to small perturbations, such as mechanical stress or alloying. Understanding the nature of the ferroelectric transition in SrTiO₃ is important as it can play an important role in other properties. For example, recently, it was found that the superconducting transition temperature is doubled in compressively strained SrTiO₃ films [1,2] for which a ferroelectric transition precedes the superconducting state upon cooling. This points to the relationship between superconductivity and ferroelectricity where ferroelectricity of SrTiO₃ can potentially be used to tune the nature of its superconducting state. SrTiO₃ is believed to exhibit a prototype displacive, soft mode ferroelectric transition. A hallmark of order-disorder transitions, on the other hand, is the formation of polar domains above the Curie temperature. These polar regions percolate below the Curie temperature to form a longrange ordered ferroelectric state. Images of polar regions above the Curie temperature of ferroelectric SrTiO₃ would provide the most direct evidence of an order-disorder transition. Atomic resolution highangle annual dark-field imaging in scanning transmission electron microscopy (HAADF-STEM) is a powerful technique that is capable of determining the atomic column positions with picometer precision and, consequently, can detect regions with subtle structural changes. Obtaining information on the columns with off-centered Ti columns from their non-polar positions sets up the premise for understanding the nature of the ferroelectric phase transition of epitaxially strained SrTiO₃.

In this study, we use HAADF-STEM to measure Ti column displacement vectors in the compressively strained SrTiO₃ films, which undergo a ferroelectric transition below 140 K. SrTiO₃ thin films were grown using hybrid molecular beam epitaxy (MBE) on (001) LSAT [(LaAlO₃)_{0.3}(Sr₂AlTaO₆)_{0.7}] and (001) SrTiO₃ single crystals. To investigate the effect of Sm doping on the ferroelectricity of strained SrTiO₃, we also studied the films with different carrier densities (n_{3D} = 6×10¹⁹, 1.4×10²⁰ and 2.8×10²⁰ cm⁻³). The polarization vector was defined as the difference between the center of mass of the four surrounding Sr columns and the actual Ti-O column position obtained by a 2D Gaussian fitting.

Figure 1 highlights the formation of the polar domain in the paraelectric phase of the compressively strained SrTiO₃ films, which undergoes a ferroelectric transition at the low temperature. Fig.1 (a-c) shows the representative HAADF-STEM image of SrTiO₃ thin films on the SrTiO₃ substrate and Fig. 1 (d-f) represents the images of strained SrTiO₃ thin films grown on LSAT substrate. The magnitudes of the polarization vector for unstrained and strained SrTiO₃ are depicted in Fig. 1(b) and (e) respectively. The arrows represent the directions of the polarization vectors and their magnitudes indicated by the color scale. The displacements in unstrained films are very small are a measure of experimental noise; in contrast, the displacements in the strained film are considerably large. Fig 1(c) and (f) show the direction of the displacement vector of the same region overlaid with the HAADF-STEM images for the un-strained and strained SrTiO₃ films respectively. The color scale is based on the polarization direction with 30-degree intervals. Whilst the Ti column displacement is random for the unstrained films, the Ti columns



in the strained SrTiO₃ film tend to displace in a favorable direction and hence form a nanodomain with the polarization direction of [001]. Figure 2 highlights the effect of doping and free charge carriers on the formation of polar domains in the strained and doped films. Sm dopants cause local lattice distortion in SrTiO₃ films [3]. Color contour maps of Sm doped samples show the increase in lattice distortion caused by the Sm dopant (~0.5% (a), ~1%(b) and ~2%(c)). The results show that the complex structural distortion caused by the Sm dopant atom shrinks the polar region size and leads to complete random distortion at a high concentration of Sm dopants [Fig 2(d) to (f)]. The highest doped SrTiO₃ film (c and f) does not undergo the ferroelectric transition at low temperature, emphasizing the importance of the formation of the high-temperature ordered phase for the emergence of the ferroelectricity [4].

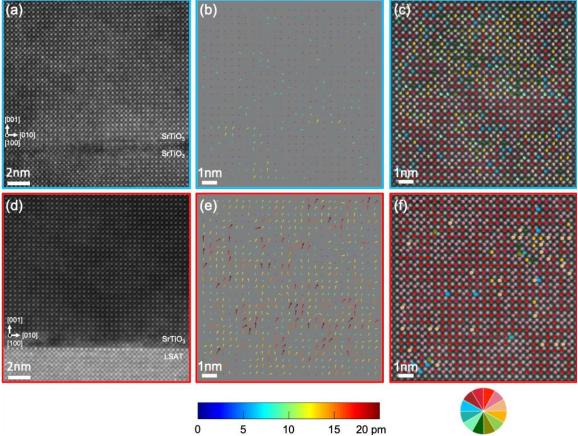


Figure 1. HAADF-STEM image of (a) unstrained and (b) strained SrTiO3. Polarization vectors for unstrained SrTiO3 (b, c) and strained (e, f) SrTiO3. The arrows in a and e indicate the magnitude of polarization. The images in the c and f display the direction of the polarization with each color corresponding to a 30-degree interval overlaid on the HAADF image

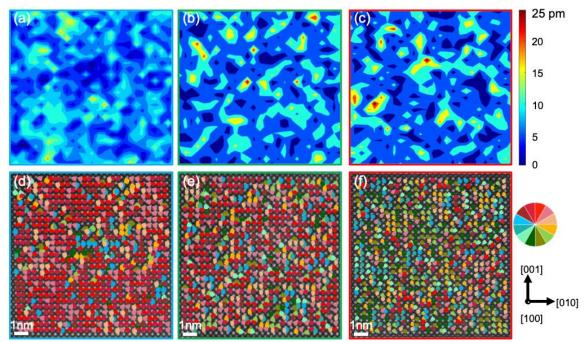


Figure 2. Color contour map (top) and polarization direction (bottom) for the SrTiO3 films with about 0.5% Sm dopant (a and d), 1% Sm dopant (b and e), and 2% Sm dopant (c and f)

References

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- [4] The authors acknowledge support by the U.S. Department of Energy (Grant No. DEFG02-2ER45994)